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1 New shock microstructures in titanite (CaTiSiO₅) from the peak ring
2 of the Chicxulub impact structure, Mexico

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29 **ABSTRACT**

30 Accessory mineral geochronometers such as apatite, baddeleyite, monazite, xenotime
31 and zircon are increasingly being recognized for their ability to preserve diagnostic
32 microstructural evidence of hypervelocity-impact processes. To date, little is known
33 about the response of titanite to shock metamorphism, even though it is a widespread
34 accessory phase and a U-Pb geochronometer. Here we report two new mechanical
35 twin modes in titanite within shocked granitoid from the Chicxulub impact structure,
36 Mexico. Titanite grains in the newly acquired core from the International Ocean
37 Discovery Program Hole M0077A preserve multiple sets of polysynthetic twins, most
38 commonly with composition planes $(K_1) = \sim\{\bar{1}11\}$, and shear direction $(\eta_1) = \langle 110 \rangle$,
39 and less commonly with the mode $K_1 = \{130\}$, $\eta_1 = \sim\langle 522 \rangle$. In some grains, $\{130\}$
40 deformation bands have formed concurrently with the deformation twins, indicating
41 dislocation slip with Burgers vector $\mathbf{b} = \langle 341 \rangle$ can be active during impact
42 metamorphism. Titanite twins in the modes described here have not been reported
43 from endogenically-deformed rocks; we therefore propose that these newly identified
44 twin form as a result of shock deformation. Formation conditions of the twins has not
45 been experimentally calibrated, and is here empirically constrained by the presence of
46 planar deformation features in quartz (12 ± 5 and $\sim 17 \pm 5$ GPa) and the absence of
47 shock twins in zircon (< 20 GPa). While the lower threshold of titanite twin formation
48 remains poorly-constrained, identification of these twins highlight the utility of
49 titanite as a shock indicator over the pressure range between 12-17 GPa. Given the
50 challenges to find diagnostic indicators of shock metamorphism to identify both

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ancient and recent impact evidence on Earth, microstructural analysis of titanite is
here demonstrated to provide a new tool for recognizing impact deformation in rocks
where other impact evidence may be erased, altered, or did not manifest due to
generally low (<20 GPa) shock pressure.

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KEYWORDS: titanite, shock metamorphism, mechanical twinning, dislocation slip
system, meteorite impact, EBSD

INTRODUCTION

Identifying and dating impacts is essential to understand their role in key processes in
Earth history, such as mass extinctions (e.g. Hildebrand et al. 1991) and even the
proposed relation to the onset of plate tectonics (O'Neill et al. 2017). Recent
crystallographic and microstructural studies of datable accessory minerals like zircon
(e.g. Moser et al. 2011; Timms et al. 2012; Erickson et al. 2013; Cavosie et al. 2016a;
Cavosie et al. 2018a; Crow et al. 2018), monazite (e.g. Erickson et al. 2016; Erickson
et al. 2017a), baddeleyite (e.g. Darling et al. 2016; Timms et al. 2017a; White et al.
2018), and xenotime (Cavosie et al. 2016b) have greatly expanded the potential to
both recognize and date impact events (e.g. Erickson et al. 2017b). Titanite
(CaTiSiO₅) is a widely-utilized U-Pb geo- and thermochronometer that occurs in a
broad range of potential target rock compositions (Frost et al. 2001), including
metamorphic, sedimentary, and felsic to mafic igneous rocks. We report a detailed
electron backscatter diffraction (EBSD) microstructural study of impact deformation
in titanite, focusing on samples of shocked granitoid from the 2016 International
Ocean Discovery Program (IODP)-International Continental scientific Drilling
Program (ICDP) Expedition 364, that drilled the peak ring of the 200 km wide
Chicxulub impact structure off the coast of the Yucatán Peninsula (Morgan et al.

2017). We further demonstrate how these features can be distinguished from deformation of titanite due to endogenic tectonic stresses.

Titanite crystallography, microstructure, and deformation

Titanite, formerly known as sphene until renamed by the International Mineralogical Association (Hey 1982), has the nominal formula CaTiSiO_5 , with an atomic structure comprising chains of octahedral 6-fold Ti and 4-fold Si, with Ca occupying a seven-fold site (Higgins and Ribbe 1976; Speer and Gibbs 1976). At ambient pressure and temperature, titanite is monoclinic, and belongs to the space group family $P2_1/a$ (Speer and Gibbs 1976; Taylor and Brown 1976). Above 496 °K (223 °C), displacements of adjacent octahedral Ti chains define a transition to an intermediate structure (Kunz et al. 1996). Above 825 °K (552 °C), or above ~3.5 GPa at room temperature, further modification of the structure results in an orthorhombic $A2/a$ symmetry that accommodates a volume reduction of 5.1% (Salje et al. 1993; Kunz et al. 1996; Angel et al. 1999). Titanite melts at >1380 °C (Hayward and Cecchetto 1982).

Titanite has good cleavage along $\{110\}$, and can form simple growth twins parallel to (100) (Deer et al. 1982). Two sets of symmetrically equivalent, polysynthetic lamellar mechanical (deformation) twins in titanite were first reported by Mügge (1889), and have since been documented in rocks affected by high-strain rates. Deformation twins were reported in samples affected by 0.5-0.8 GPa pressure during nuclear tests (Borg 1970), and were subsequently produced in laboratory experiments at 0.8 GPa (Borg and Heard 1972). The twin mode for deformation twins was defined as $K_1 = \sim\{221\}$, $\eta_1 = \langle 110 \rangle$, where K_1 = twinning (or composition) plane and η_1 = shear direction (Borg 1970). With a critically resolved shear stress of ~0.13

102 GPa at 500 °C, $\sim\{221\}\langle 110\rangle$ twinning can occur in titanite at typical endogenic
 103 tectono-metamorphic conditions (Borg and Heard 1972; Bonamici et al. 2015).
 104 Titanite can also undergo dislocation creep (Müller and Franz 2004; Bonamici
 105 et al. 2015) and dynamic recrystallization (Papapavlou et al. 2017) in tectono-
 106 metamorphic settings. Crystal-plastic deformation of titanite is not well studied.
 107 Dislocations with Burgers vectors (**b**) = [100], [011] and $[0\bar{1}1]$ have been reported in
 108 titanite at eclogite facies, and easy glide of dislocations with **b** = $\frac{1}{2}\langle 011\rangle$ that is
 109 predicted to occur in the A2/a phase (Müller and Franz 2004). Recrystallised
 110 subgrains with low- and high-angle boundaries have also been reported in
 111 tectonically-deformed titanite (Bonamici et al. 2015; Papapavlou et al. 2017; Kirkland
 112 et al. 2018).
 113 Titanite grains have been experimentally dynamically shocked to 59 GPa
 114 (Deutsch and Schärer 1990). However, the resulting microstructures were not
 115 characterised in detail. Previously reported titanite microstructures in naturally
 116 shocked rocks that did not involve quantitative approaches have been described
 117 simply as ‘subplanar fractures’ (e.g. Koeberl et al. 1996; Papapavlou et al. 2018),
 118 ‘planar fractures’ and ‘mechanical twinning’ (Biren and Spray 2011). In breccia from
 119 the Ries crater, Germany, Chao (1968) reported weakly-developed planar features and
 120 misoriented mosaic textures. Whereas Abadian (1972) reported dominant cleavage
 121 along (111) and (110), less common cleavage in an irrational plane $\parallel(552)$, and fine
 122 (1-2 μm wide) planar elements parallel to the {010} zones. Electron backscatter
 123 diffraction analysis of titanite in target rocks from the Sudbury impact structure,
 124 Canada, and Vredefort impact structure, South Africa, have been shown to be crystal-
 125 plastically deformed, and to contain lamellar twins with $\sim 74^\circ / \langle 102\rangle$ host-twin
 126 disorientation axes, and neoblasts (Papapavlou et al. 2018). However, twins identified

in these studies have not been indexed. Until now, no studies have focused on quantitatively distinguishing the style of deformation features that form in titanite in tectonic stress regimes from those that form during impact events, which is the focus of this study.

Geological background and samples

This study investigates titanite from the ~66 Ma, 200 km Chicxulub impact structure in the Gulf of Mexico, which is widely reported to have caused the K-Pg mass extinction (Hildebrand et al. 1991; Schulte et al. 2010; Renne et al. 2013). It is one of the three largest known terrestrial impact structures (Grieve and Theriault 2000) and contains a well-preserved peak ring (Fig. 1) (Morgan et al. 1997; Gulick et al. 2008; Gulick et al. 2013; Morgan et al. 2016; Riller et al. 2018). In 2016, IODP-ICDP Expedition 364 drilled core from Hole M0077A (21° 27.009'N, 89° 56.962'W), which penetrated the peak ring (Morgan et al. 2017) (Fig. 1). Recovered lithologies include a section of shocked Paleozoic granitoid basement rocks from depths of 747-1335 meters below sea floor (mbsf) that contain pre-impact mafic and felsic dike lithologies as well as intercalations of impact breccia and impact melt rocks (Fig. 1) (Morgan et al. 2017).

The granitic target rocks are coarse-grained and primarily composed of alkali feldspar, quartz, plagioclase, minor biotite, and accessory muscovite, apatite, titanite, epidote, magnetite, ilmenite, and zircon (Gulick et al. 2017; Morgan et al. 2017; Schmieder et al. 2017). Most quartz grains contain multiple decorated planar deformation features (PDFs), planar fractures (PFs), and feather features (FFs) (Fig. 2) (e.g. Ferrière et al. 2017; Gulick et al. 2017; Morgan et al. 2017; Rae et al. 2017; Zhao et al. 2017; Rae 2018). Preliminary universal stage (U-stage) analysis of PDF

orientations in shocked quartz constrains the bulk shock pressure of rocks in the core to $\sim 12 \pm 5$ to 17 ± 5 GPa (Rae et al. 2017; Feignon et al. 2018; Rae 2018). Titanite grains up to 2 mm long occur as inclusions within all major phases of the granitic rocks, including shocked quartz, and are often spatially associated with and/or include other accessory phases, such as zircon and apatite (Figs. 2, 3).

ANALYTICAL METHODS

Samples and approach used in this study

Titanite grains and associated phases were characterised via optical microscopy, backscattered electron (BSE) imaging, and EBSD mapping in thin sections of three shocked granite core samples from Hole M0077A (364-77-A-121-R-1-75-77, 364-77-A-204-R-1-7-9, and 364-77-A-219-R-1-22-24) from depths of 814.85, 1030.00, and 1076.16 metres below sea floor (mbsf), respectively (Figs. 2, 3, and Table 1). The samples are referred to throughout this study by these depths with # prefix (Table 1). Modes of twinning in titanite identified from EBSD data were determined via analysis of the host-twin crystallographic orientation relationships. A MATLAB script was used to determine the directions defining 180° misorientation relationships, which correspond to either the twinning direction of shear (η_1) or the pole to the compositional plane (twin plane, K_1) (Christian and Mahajan 1995; Erickson et al. 2016). Finally, twins were indexed using geometric considerations (Christian and Mahajan 1995) combined with the traces of twin lamellae on EBSD maps. The geometry of low-angle lattice distortions associated with deformation bands seen in EBSD data were analysed, including automated calculation of the weighted Burgers vector (Wheeler et al. 2009).

177 **Scanning electron microscopy and electron backscatter diffraction mapping**

178 Prior to scanning electron microscopy, thin sections were polished progressively with
179 diamond paste to 0.5 μm , then given a final polish using 0.06 μm colloidal silica in
180 NaOH on a Buehler Vibromet II for four hours. A thin carbon coat was applied to
181 mitigate charging. Backscattered electron images were collected using a Tescan
182 MIRA3 field emission scanning electron microscope (FE-SEM) in the John de Laeter
183 Centre at Curtin University. Phases and their crystallographic microstructures were
184 quantified using combined energy dispersive X-ray (EDX) and electron backscatter
185 diffraction (EBSD) mapping using an Oxford Instruments AZtec system on the
186 Tescan MIRA3 FE-SEM at Curtin University. Data acquisition settings and
187 processing procedures followed those of Timms et al. (2017b) and are detailed in
188 Table 2. EBSD data were processed using the Tango and Mambo modules of Oxford
189 Instruments Channel 5.10 to produce thematic maps and pole figures, respectively.

190

191 **Determining twin modes from EBSD data**

192 EBSD mapping involves quantification of the crystallographic orientation of
193 phases relative to the sample surface. It is conventional to describe this absolute
194 crystallographic orientation in the x-y-z sample reference frame at a given x-y point
195 on the map as a sequence of three rotations from a reference orientation, known as
196 Euler angles ϕ_1 , Φ , and ϕ_3 (Bunge 1981; Prior et al. 1999). Twins are recognised in
197 EBSD maps as domains with a specific systematic misorientation relationship relative
198 to the host grain. Misorientation is defined as the rotation around an axis by some
199 angle that would bring any two differently orientated crystals into alignment (Wheeler
200 et al. 2001). Due to the symmetry of crystals, multiple misorientation axes are present.
201 Each axis corresponds to a different angular rotation, and all of these orientations

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202 describe the relationship between the two domains. The most commonly reported type
203 of misorientation in EBSD data sets is the axis around which the minimum rotation is
204 required for realignment, which is specified as the disorientation (Wheeler et al.
205 2001). This distinction is important when considering mechanical twinning, where it
206 is conventional to classify twin modes based on rotations around axes where the
207 misorientation angle is 180° , and which may be different to the disorientation.

208 Deformation twins are characterized using one or both of the twinning /
209 composition plane, K_1 , and shear direction, η_1 , which results in 180° misorientation
210 relationships (Christian and Mahajan 1995). Deformation twins can be classified in
211 one of three ways. Type 1 twins have a rational K_1 , the pole to which is a
212 misorientation axis with rotation angle of 180° , and an irrational η_1 . Conversely, type
213 2 twins have rational η_1 , around which is a misorientation axis with the rotation angle
214 of 180° , and an irrational K_1 , and compound twins are defined by both rational K_1 and
215 η_1 , both of which are misorientation axes with rotations of 180° (Christian and
216 Mahajan 1995). In general, the relationship reported by EBSD software will not have
217 a rotation angle of 180° , as there will usually be a lower angle disorientation
218 relationship. The exceptions to this issue are in triclinic minerals (where there is only
219 one symmetric equivalent) and compound twins in monoclinic minerals (where there
220 are two symmetric equivalents and therefore they must both be components of the
221 twinning). This relationship means it is possible for the disorientation angle/axis to
222 bear no resemblance to K_1 and η_1 for various twin modes in most crystal systems.

223 A more sophisticated analysis of EBSD data that involved calculating 180°
224 misorientations is required to identify K_1 and η_1 for twinning in monoclinic phases
225 such as monazite (Erickson et al. 2016). A MATLAB script was developed that
226 utilizes Euler angle triplets for two representative data points from the host and

adjacent twin domain derived from the EBSD map, crystal unit cell parameters, and Laue group symmetry operators as input parameters (Table 3).

First, the rotation matrix (\mathbf{R}), which describes the orientation relative to a reference orientation, is calculated for each Euler triple (host \mathbf{R}_h , twin \mathbf{R}_t) using equations B2 and B4 of Cho et al. (2005). These can be used to calculate the directions in sample coordinates of particular crystal directions. For our purposes we require a matrix which describes the rotation from host to twin *expressed in crystal coordinates* ($\mathbf{R}_m = \mathbf{R}_h^{-1}\mathbf{R}_t$). Since this is in crystal coordinates, we derive rotation matrices for each symmetric equivalent ($\mathbf{R}_m i = \mathbf{S}_i \mathbf{R}_m$) where \mathbf{S}_i is the i -th member of the set of symmetry operators for the Laue class under consideration. The 180° misorientation axes were calculated as angle/axis pairs in Cartesian x-y-z coordinates (the sample reference frame). These were used to find best-fit crystallographic forms using the dot product of unit vectors, as $\langle hkl \rangle$ or pole to $\{hkl\}$, implementing integer search limits of $\bar{4}$ to 4 for h , k , and l to yield low-index (rational) best-fit forms. Output data were compiled giving the crystallographic vector with the smallest value normalised to 1 and their corresponding angular deviation from the calculated misorientation axis. The orientation of K_1 , η_1 and S were reconstructed using a stereographic projection, ensuring that the appropriate symmetric variants of K_1 and η_1 were chosen such that its great circle contains the mapped x-y trace of the twins and η_1 .

The K_1 and η_1 components of twin modes were deemed to be rational if angular deviation between the calculated misorientation axis and the best-fit low index crystallographic form were $<0.7^\circ$, which is a reasonable value given the mean angular deviation associated with indexing EBSD data points is typically on the order of 0.3 to 0.8° . Twin mode components were considered irrational where angular

deviation to the best-fit low-index form were $>0.7^\circ$ (typically $>1.5^\circ$). These results were used to classify twin modes as types 1, 2, or compound twins, as described above (Christian and Mahajan 1995). Results are summarized in Tables 4 and 5. For a given twin mode, the disorientation axis is assumed to be defined by the intersection of planes normal to η_1 for both symmetrically equivalent sets of twins. Therefore, the disorientation axis associated with $\sim\{221\} \langle 110 \rangle$ twins (Borg 1970; Borg and Heard 1972) is parallel to $[102]$, which is coincident with the pole to (001) .

Determination of slip systems

Two approaches were used to determine dislocation slip systems active in shocked titanite. The first approach involved a geometric analysis of crystal-plastic deformation bands using EBSD data, assuming a simple tilt-boundary model, which has been successfully applied to low-angle boundaries in other minerals (Boyle et al. 1998; Bestmann and Prior 2003; Reddy et al. 2007). This approach assumes that disorientation axes associated with low-angle deformation bands are a consequence of geometrically necessary dislocations, and that the disorientation axis, pole to the slip plane, and Burgers vector (**b**) are orthonormal. Geometric reconstruction of low-angle tilt boundary planes must contain their traces on EBSD maps and the disorientation axis. The **b** is oriented normal to the tilt boundary, and the slip plane is assumed to contain both **b** and the disorientation axis. The second approach involved automated calculation of the weighted Burgers vector (WBV) of geometrically necessary dislocations using orientation gradients in EBSD data (Wheeler et al. 2009). The WBV is defined as the net Burgers vector of dislocations that intersect an area of the EBSD map, and has been calculated by integration around the edge of a user-defined area.

277 In this study, an automated version of the integral approach of Wheeler et al.
278 (2009) was implemented, whereby an EBSD map is automatically tiled into square,
279 20 x 20 pixel (6 x 6 μm) areas, and the WBV calculated for each tile. This method
280 reduces the errors on WBV in comparison to values calculated at every pixel. Results
281 from tiles were disregarded if the square overlapped a high-angle boundary (including
282 twin interfaces) or where the magnitude of the WBV length was below a threshold
283 value (i.e., a minimum dislocation density, $\text{ddmin} = 0.001 \text{ } (\mu\text{m})^{-1}$). The two
284 approaches to deducing dislocation slip systems described above are complementary:
285 the WBV approach is automated and does not involve assumptions about low-angle
286 boundary geometry, whereas the low-angle boundary model also allows indexing of
287 low-angle deformation bands.

288

289 **RESULTS**

290 **Microstructures in Chicxulub Titanite**

291 Four titanite grains were analysed, including a >1 mm long grain in #814.85; a
292 200 μm grain in #1030.00; and two grains from #1076.16 that are 300 μm and 500 μm
293 across, respectively (Figs. 2, 3). The titanite grains occur as inclusions within shocked
294 and fractured alkali feldspar and quartz, and are cut by brittle fractures (Figs. 2-7).
295 None of the studied grains occur in discrete breccia, melt, or cataclasite veins, and are
296 entirely enclosed within granitoid (Fig. 2). Surrounding quartz grains contain multiple
297 sets of PDFs, PFs, FFs (Fig. 2), and high densities of lobate/irregular Dauphiné twins
298 (Figs. 6, 7). Calcite and TiO_2 are commonly present along fractures in titanite and
299 along grain boundaries (Figs. 3-7).

300 All titanite grains were indexed as $\text{P2}_1/\text{a}$, with no evidence of the A2/a
301 structure. All four titanite grains contain two or more sets of polysynthetic twin

lamellae (Figs. 4-7). Twin lamellae are typically a few micrometres wide, straight to slightly kinked, and tapered. Others terminate against grain boundaries, fractures, or other twins, and are unevenly developed across the grains (Figs. 4-7). In some grains, conspicuous sub-planar partings are present along twin interfaces (e.g., Figs. 4, 5). Each titanite grain contains two sets of twins that are disorientated from the host grain by $\sim 74^\circ$ / [102] (labelled T1 and T2 in Figs. 4-7). The grain from sample #1030.00 contains a third set of twins with a disorientation relative to the host of $\sim 51^\circ$ / [001] (labelled T3 in Fig. 7A).

All host domains in the titanite grains record variable degrees of intragrain misorientation, up to $\sim 30^\circ$, which manifests as dispersion in pole figures (Figs. 4-7). While some dispersion is attributed to rigid rotation of fractured blocks, a component is linked to progressive distortion and broad deformation bands with indistinct boundaries, which are the result of crystal-plasticity (e.g., Fig. 5). Significant thickening of twins along deformation bands in titanite grain 1 in #1076.16 indicates that the formation of these microstructures was coeval (Fig. 8). The high angle of the twin planes to the polished surface for this sample (Fig. 8B) means that the true thickness is observed, and the twin thickness is not merely an apparent thickness, due to the plane of observation. The best-developed (strongest and most planar) deformation bands in the analysed titanite grains are defined by systematic crystallographic disorientation about an axis sub-parallel with the pole to {111} (Fig. 8A-B). The map traces of the deformation bands are geometrically consistent with low-angle tilt boundaries along {130} that contain the disorientation axis (Fig. 8B). This geometric configuration can be explained by dislocation glide with Burgers vector $\mathbf{b} = [341]$. This result is supported by weighted Burgers vector analysis, which

shows WBV forms dominant clusters around [341] (Fig. 8C-E), which has not previously been described in titanite.

Indexing of Twins in Titanite

The most common twins encountered in the Chicxulub titanite grains have disorientation relative to the host of $\sim 74^\circ$ / [102], and 180° misorientation axes that align with $\langle 110 \rangle$ of the host grains, which is coincidental with η_1 of twins described by Borg (1970) (Twins 1 and 2 in Figs. 4-7; Table 5). However, reconstruction of the composition plane (K_1) using the map trace of these twins is consistent within a few degrees of $\{\bar{1}11\}$, rather than $\{221\}$ expected for Borg (1970) twins. Therefore, they define two symmetric equivalents of a previously undescribed type 2 twin mode with $K_1 =$ irrational $\sim\{\bar{1}11\}$, $\eta_1 =$ rational $\langle 110 \rangle$, and shear plane (S) = $\{112\}$ (Fig. 10, Table 5). A geometric consequence of this host-twin orientation relationship is that it results in coincidence of many crystallographic planes between twin and host that lie normal to $\eta_1 \langle 110 \rangle$, the 180° misorientation axis (Figs. 4-7, Table 5).

Twins with $\sim 51^\circ$ / [001] disorientation were found in one grain (labelled T3 in Fig. 7A). The 180° misorientation axis for these twins aligns with the pole to $\{130\}$, which defines a previously undescribed twin mode $K_1 = \{130\}$, $\eta_1 =$ irrational $\sim\langle 522 \rangle$, and $S = (10\bar{3})$ (Table 5; Fig. 7B, 10). Two symmetrically equivalent variants are possible for this twin mode, which are classified as type 1 twins because they have a rational K_1 (Christian and Mahajan 1995) (Fig. 10).

Distinguishing between the newly described type 2 $\sim\{\bar{1}11\}$ twins and the established $\sim\{221\}$ twins requires care, because they have identical η_1 (parallel to $\langle 110 \rangle$) and thus produce identical disorientation axis/angle relationships using EBSD data (Fig. 10, Table 5). Therefore, correct indexing requires consideration of the trace

of twin planes on the EBSD map and η_1 to reconstruct K_1 (e.g., Figs. 4-7). Indexing of the newly described type 1 $\{130\} \sim \langle 522 \rangle$ twins is straightforward because it can be inferred from the disorientation angle/axis alone, as no other twins are known to have similar relationships.

Microstructures in Co-existing Zircon

A total of eight zircon grains were analysed in the same thin sections as the titanite grains, including zircon grains that occur as inclusions within the larger shocked titanite grains (Fig 3A). The zircon grains are typically $<60 \mu\text{m}$ across, euhedral to anhedral in shape, with some grains preserving evidence of growth zoning. All of the grains are fractured, and most of the observed variations in crystallographic orientation, shown by variations in colour in EBSD orientation maps and pole figures in Fig. 9, are related to rotation of rigid blocks separated by fractures (Fig. 9A, B, D, F, and G). However, the zircon grains preserve minor evidence ($<5^\circ$) of crystal-plastic strain, seen as smooth, systematic gradients in orientation that are not related to fractures (Fig. 9A, E, and H). No microstructures diagnostic of shock metamorphism, such as $\{112\}$ twins, the high-pressure polymorph reidite, granular neoblasts, or dissociation textures (e.g. Timms et al. 2017b), were detected in the zircon grains analysed. If present, these microstructures would have been readily detected by the EBSD analysis utilized in this study.

DISCUSSION

New titanite twins in Chicxulub shocked granite

The two twin types found in Chicxulub titanite described here have not been reported previously in tectonically-deformed titanite, and we propose that they are a

376 product of shock metamorphism during hypervelocity impact conditions given the
 377 occurrence in granitoid with well-documented quartz microstructures indicative of
 378 shock pressures of about 12 to 17 GPa. Thus, our results offer new insights into how
 379 titanite deforms in impact environments. The new twin modes are geometrically
 380 distinct from previously established mechanical and/or growth twins in titanite (e.g.
 381 Borg 1970) (Fig. 10, Tables 4 and 5). The most common twins in Chicxulub titanite
 382 grains have irrational composition planes (K_1) that are within a few degrees of $\{\bar{1}11\}$,
 383 and a rational $\eta_1 = \langle 110 \rangle$, and $(S) = \{112\}$. Chicxulub titanite grains typically have
 384 two sets of $\sim\{\bar{1}11\} \langle 110 \rangle$ twins, which is the maximum number possible given the
 385 monoclinic symmetry. One unusual aspect of $\sim\{\bar{1}11\} \langle 110 \rangle$ twins is that they share
 386 an identical twin-host minimum misorientation (disorientation) relationship with
 387 previously established $\sim\{221\} \langle 110 \rangle$ twins (e.g., Borg, 1970), and thus the two twin
 388 types cannot be distinguished solely based on disorientation angle/axis measurements
 389 provided by EBSD analysis. Identification of each twin type thus requires indexing of
 390 the composition planes (twin planes), which we calculated using a MATLAB script.

391 A set of lamellar twins with a different host-twin disorientation
 392 crystallographic relationship of $\sim 51^\circ / [001]$ was found in one Chicxulub grain
 393 (labelled T3 in Fig. 2A). These twins represent a second previously undescribed twin
 394 mode in titanite, whereby $K_1 = \{130\}$, $\eta_1 = \text{irrational } \sim\langle 522 \rangle$, and $S = (10\bar{3})$ (Table
 395 4; Fig. 7B, 9C). Indexing of the newly described $\{130\} \sim\langle 522 \rangle$ twins is
 396 straightforward because it can be inferred from the disorientation angle/axis relations
 397 provided by EBSD data.

398

399 **Concurrent deformation processes in titanite during shock metamorphism**

400 The thickening of $\sim\{\bar{1}11\} \langle 110 \rangle$ mechanical twins along $\{130\}$ deformation
 401 bands clearly indicates that dislocations with \mathbf{b} $[341]$ occurred concurrently with
 402 twinning (Fig. 8). Concurrent crystal-plasticity could potentially have affected
 403 twinning in two different ways: (1) the widening of twins was mechanically
 404 facilitated by $\{\bar{1}12\} [341]$ dislocation activity, and/or (2) the formation of
 405 deformation bands locally re-orientated titanite into a more favourable orientation for
 406 twinning with respect to the stress field of the shock wave. Given that the slip vector
 407 for twinning and Burgers vector for dislocation glide are similarly-oriented (i.e.,
 408 twinning $\eta_1 = \langle 110 \rangle$ and dislocation $\mathbf{b} = \langle 341 \rangle$, respectively, are within $\sim 10^\circ$ of one
 409 another), it is possible that these two types of microstructure have accommodated
 410 shock deformation in a coherent and systematic way similar to cross slip. However,
 411 the effects of localised re-orientation in deformation bands on the ease of twinning
 412 cannot be resolved without additional information about the nature of the stress-strain
 413 field associated with the passing shock wave relative to the orientation of the grains
 414 and critically-resolved shear stress for twinning. Nevertheless, the absence of $\{130\}$
 415 deformation bands in other shock-twinned grains in this study indicates that
 416 concurrent $[341]$ dislocation slip is not a requirement for these twin modes in titanite.

418 **Petrological implications of the new twin modes in titanite**

419 Titanite in the target rocks that formed the peak ring at the Chicxulub crater
 420 responded to the $\sim 12\text{--}17$ GPa bulk peak shock pressure principally by $\sim\{\bar{1}11\} \langle 110 \rangle$
 421 twinning, with minor $\{130\} \sim\langle 522 \rangle$ twinning and coeval crystal-plasticity. Zircon, a
 422 tetragonal accessory phase, is also known to form deformation twins and other
 423 microstructures during shock deformation (Fig. 11). In contrast to titanite, zircon
 424 grains in the same samples do not record diagnostic shock/impact-related

microstructures (Fig. 9), which is consistent with their having experienced shock pressure <20 GPa (e.g. Timms et al. 2017b). In this respect, titanite appears similar to xenotime, which has also been shown by empirical calibration with quartz and zircon to form impact-related deformation twins and plastic deformation at shock pressures <20 GPa (Cavosie et al. 2016b) (Fig. 11). Differences in the response of titanite and other accessory minerals to shock deformation are shown in Fig. 11. The variable microstructural responses are presumably related to the intrinsic material properties of the different minerals, such as elasticity and yield strength for various failure modes, which are largely controlled by crystal structure. Phase transformations and reactions that determine phase stability through shock conditions are also important.

The precise details of the kinetics, nucleation stress, and critically resolved shear stress for the newly described twin modes are yet to be determined by theoretical or experimental means. However, unlike for shock twinning and dislocations in zircon (Timms et al. 2018), theoretical calculations are inhibited by the current lack of published elastic constants for P2₁/a titanite. More rigorous investigations into the shock response of titanite via laboratory shock deformation experiments remain an avenue for future research. Furthermore, very few quantitative microstructural studies of naturally deformed titanite from tectonic and/or impact environments are currently available (Papapavlou et al. 2017; Papapavlou et al. 2018). Thus, further studies of naturally shocked and tectonically-deformed titanite are required to provide better constraints on formation conditions for different twin modes.

The discovery of new titanite twin modes in shocked target rocks at Chicxulub represents the first steps toward developing a twin-based framework for using titanite to distinguish tectonic versus impact-related deformation, similar to the approach

developed recently for monazite (Erickson et al. 2016). Titanite appears to behave similarly to monazite in that a range of twin modes have been reported, with empirical studies indicating that certain twin types uniquely form as a consequence of shock deformation. Cleavage along $\sim\{\bar{1}11\}$ twin planes shown by several of the Chicxulub grains is different to the dominant cleavage orientations reported for titanite from the Ries crater (Abadian 1972). If titanite commonly cleaves along twin lamellae, then observations of cleavage made by Abadian (1972) may be an indication that additional shock twin modes could be revealed via detailed studies of shocked titanite from other impact structures (Papapavlou et al. 2018). Our findings further indicate that additional information is required to singular twin disorientation axes produced from EBSD data in order to correctly index twins in titanite.

Broader applications of deformation twin systematics

In this study, we outline an approach for rigorous indexing and classification of deformation twins from EBSD data based on geometric considerations of Christian and Mahajan (1995). We demonstrate the importance of our approach for distinguishing twin modes in titanite formed in tectonic versus shock settings. This approach builds on the method outlined by Erickson et al. (2016) that led to the discovery of new twin modes in monazite, and can be applied to other minerals that form twins, such as ilmenite, rutile, and pyrrhotite, which could be useful for better understanding deformation of rocks in both tectonic and impact settings.

CONCLUSIONS

In conclusion, titanite is a common accessory phase in a variety of rock types (Frost et al. 2001), which has the propensity to record impact-related microstructures.

Specifically, we demonstrate titanite undergoes shock-related deformation twinning along $\sim\{\bar{1}11\}$ and $\{130\}$, as recorded in the shocked target rocks of the Chicxulub impact structure over the pressure range between 12 ± 5 and $\sim 17 \pm 5$ GPa. These twin modes can form concurrently with deformation bands along $\{130\}$ resulting from dislocation migration with a $\langle 341 \rangle$ Burgers vector. The newly-described twin modes in $\sim\{\bar{1}11\}$ and $\{130\}$ are different from previously reported $\sim\{221\}$ twins from tectonically-deformed titanite, and we therefore hypothesize that they are indicative of shock conditions. We have defined geometric criteria for distinguishing these various twins in titanite, which highlights the importance of utilizing a rigorous approach for indexing twins. Furthermore, as this accessory mineral may be susceptible to age-resetting during deformation (Papapavlou et al. 2017; Papapavlou et al. 2018), we propose that our findings greatly increase the potential of titanite as a diagnostic recorder of impact events in the geological record.

488

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FIGURE CAPTIONS

Figure 1. A. Gravity anomaly map of the Chicxulub impact structure showing traces of various morphological features of the buried crater (dashed lines), location of Hole M0077A, and sinkholes (cenotes) (white dots), and present-day coastline (solid white line). B. Schematic core log of Hole M0077A showing the distribution of lithologies and locations of samples used in this study. After Morgan et al. (2016) and Gulick et al. (2013).

Figure 2. Optical photomicrographs of the samples used in this study. Mineral abbreviations after Whitney and Evans (2009) include: Qz = quartz; Afs = alkali feldspar; Pl = plagioclase; Ttn = titanite; Chl = chlorite; Bt = biotite. A. #814.85. B. #1030.00. C. #1076.16. D. Region of interest in #1076.16 shown by white box in C. White arrows annotate significant fracture orientations. Samples were impregnated with blue epoxy prior to thin sectioning to enable the porosity to be visualized. Plane polarized light images.

Figure 3. Backscattered electron images of titanite grains. Mineral abbreviations after Whitney and Evans (2009) include: Qz = quartz; Afs = alkali feldspar; Pl = plagioclase; Ttn = titanite; Zrn = zircon; Cal = calcite; Ap = apatite; Mag = magnetite;

525 Bt = biotite; Hc = hercynite. A. K-feldspar-hosted titanite in #814.85 shown in Fig.
 526 2A. B. Detail of the margin of the titanite grain in A, showing a fractured zircon grain
 527 and five sets of planar microstructures in alkali feldspar (black arrows). C. Primary
 528 titanite, magnetite and apatite in a quartz domain in #1030.00 shown in Fig. 2B. D.
 529 Apatite grains with planar fractures in an alkali feldspar domain in #1030.00. E.
 530 Titanite in an alkali feldspar domain in #1076.16 shown in Fig. 2C. F. Titanite with
 531 apatite, magnetite, zircon and hercynite in a quartz domain in #1076.16 shown in Fig.
 532 2D. White arrows in A-F indicate the trace of transgranular fractures.

534 **Figure 4.** Microstructure of titanite grain from #814.85 (Fig. 3A). A. Cumulative
 535 disorientation map showing 30° variation across the grain and two sets of twins
 536 (labelled T1 and T2). An inclusion of a titanite grain with a different orientation is
 537 shown by white outline labelled (i). B-C. Detailed maps of region shown in A. D.
 538 Pole figures for (100), (010) and (001). Lower hemisphere, equal area projections in
 539 the sample x-y-z reference frame. E. Crystallographic relationships between twin and
 540 host for twins shown in A-C in the sample reference frame.

542 **Figure 5.** Microstructure of titanite grain 1 from #1076.16 (Fig. 3E). A. Cumulative
 543 disorientation map showing 20° variation across the titanite grain and two sets of
 544 twins (labelled T1 and T2). Fractures and perthitic lamellae are present in surrounding
 545 K-feldspar grains. Inset (i) is a detailed map showing T2 twins. B. Pole figures for
 546 (100), (010) and (001). Sparse systematically misindexed points at host-twin
 547 interfaces (labelled as 's.m.') have disorientation relationships of 98° / $\langle \bar{2}\bar{1}2 \rangle$ and
 548 180° / [003]. Lower hemisphere, equal area projections in the sample x-y-z reference

frame. C. Crystallographic relationships between twin and host for twins shown in A in the sample reference frame.

Figure 6. Microstructure of titanite grain 2 from #1076.16 (Fig. 3F). A. Cumulative disorientation map showing 30° variation across the titanite grain and two sets of twins (labelled T1 and T2). Dauphiné twins are present in surrounding quartz grains, and magnetite contains thin lamellar twins. Inset (i) is a detailed map showing T2 twins. B. Pole figures for (100), (010) and (001). Lower hemisphere, equal area projections in the sample x-y-z reference frame. C. Crystallographic relationships between twin and host for twins shown in A in the sample reference frame.

Figure 7. A. EBSD map of titanite in sample #1030.00 from the Chicxulub crater peak ring (Fig. 3C). Titanite is coloured for cumulative disorientation relative to reference orientation (red cross) and has three sets of twins (T1, T2, and T3). It is surrounded by quartz with Dauphiné twins (Qz, orange/red), calcite (Cal, yellow) and TiO₂. Inset (i) shows pole figure for (100) in sample reference frame. B. Crystallographic relationships between twins T1, T2, and T3 and host grain. Several low-index poles to coincident planes (grey circles) align along the plane normal to η_1 for Twins 1 and 2. Lower hemisphere, equal area projections in the sample x-y-z reference frame.

Figure 8. A. Detailed EBSD map of grain 1 from #1076.16 shown in Fig. 5. Titanite is coloured for cumulative disorientation relative to reference orientation (red cross) and twins (purple, labelled T1). Crystal-plastic deformation band (green-orange domain) has a well-defined lower boundary trace (white dashed line), and is displaced

by brittle fractures. Twin lamellae are thicker within the deformation band. B. Composite pole figure of data shown by the white box in A, plotted in the sample x-y-z reference frame. Crystallographic data show a systematic dispersion about a disorientation axis approximately parallel to the pole to (111). The deformation band is consistent with a {130} tilt boundary geometry that contains the disorientation axis and the trend of the deformation band on the polished surface. The dominant dislocation slip system that contributed to the deformation bands involved glide with a [341] Burgers vector. C EBSD map of the entire grain showing results of automated weighted Burgers vector (WBV) analysis. Superimposed squares indicate 6 x 6 μm tiles from which the integral WBV was calculated. Tiles that overlap with twin domains have been disregarded. Tiles are coloured for WBV orientation using an inverted pole figure (IPF) colour scheme. D Pole figure of WBV data shown in C, plotted in crystallographic reference frame. E Contoured pole figure of data shown in D, indicating dominance of WBV parallel to [341].

Figure 9. Microstructure of zircon grains associated with titanite from samples #814.85 and #1030.00 of this study.

Figure 10. Summary of the geometric elements of deformation microstructures in titanite. Pole Figures A. $\sim\{221\}$ $\langle 110 \rangle$ twins after Borg (1970). B. $\sim\{\bar{1}11\}$ $\langle 110 \rangle$ twins, this study. C. $\{130\}$ $\sim\langle 522 \rangle$ twins, this study. D. $\{130\}$ deformation bands, this study. The two possible symmetric equivalent variants are shown for each type of microstructure. Microstructures shown in B, C, and D have not been described from tectonically-deformed titanite.

Figure 11. Accessory phases as indicators of shock metamorphism. [1]: Leroux et al. (1999), [2]: Timms et al. (2012), [3]: Moser et al. (2009), [4]: Timms et al. (2017a), [5]: Erickson et al. (2013), [6]: Nemchin et al. (2009), [7]: Timms et al. (2018), [8]: Moser et al. (2011), [9]: Thomson et al. (2014), [10]: Cox et al. (2018), [11]: Wittmann et al. (2006), [12]: Cavosie et al. (2015b), [13]: Reddy et al. (2015), [14]: Erickson et al. (2017a), [15]: Cavosie et al. (2015a), [16]: Cavosie et al. (2016a), [17]: Cavosie et al. (2018b), [18]: Timms et al. (2017b), [19]: Erickson et al. (2015), [20]: Erickson et al. (2016), [21]: Erickson et al. (2017b), [22]: Cavosie et al. (2016b), [23]: Darling et al. (2016), [24]: White et al. (2018), [25]: Müller and Franz (2004), [26]: Borg (1970), [27]: Papapavlou et al. (2018), [28]: Papapavlou et al. (2017). Note that the pressures indicated are specifically mean (bulk) pressures for rocks with negligible initial porosity, such as the granitoids in this study. The effects of porosity on bulk shock pressure have been treated elsewhere (e.g., Güldemeister et al. 2013).

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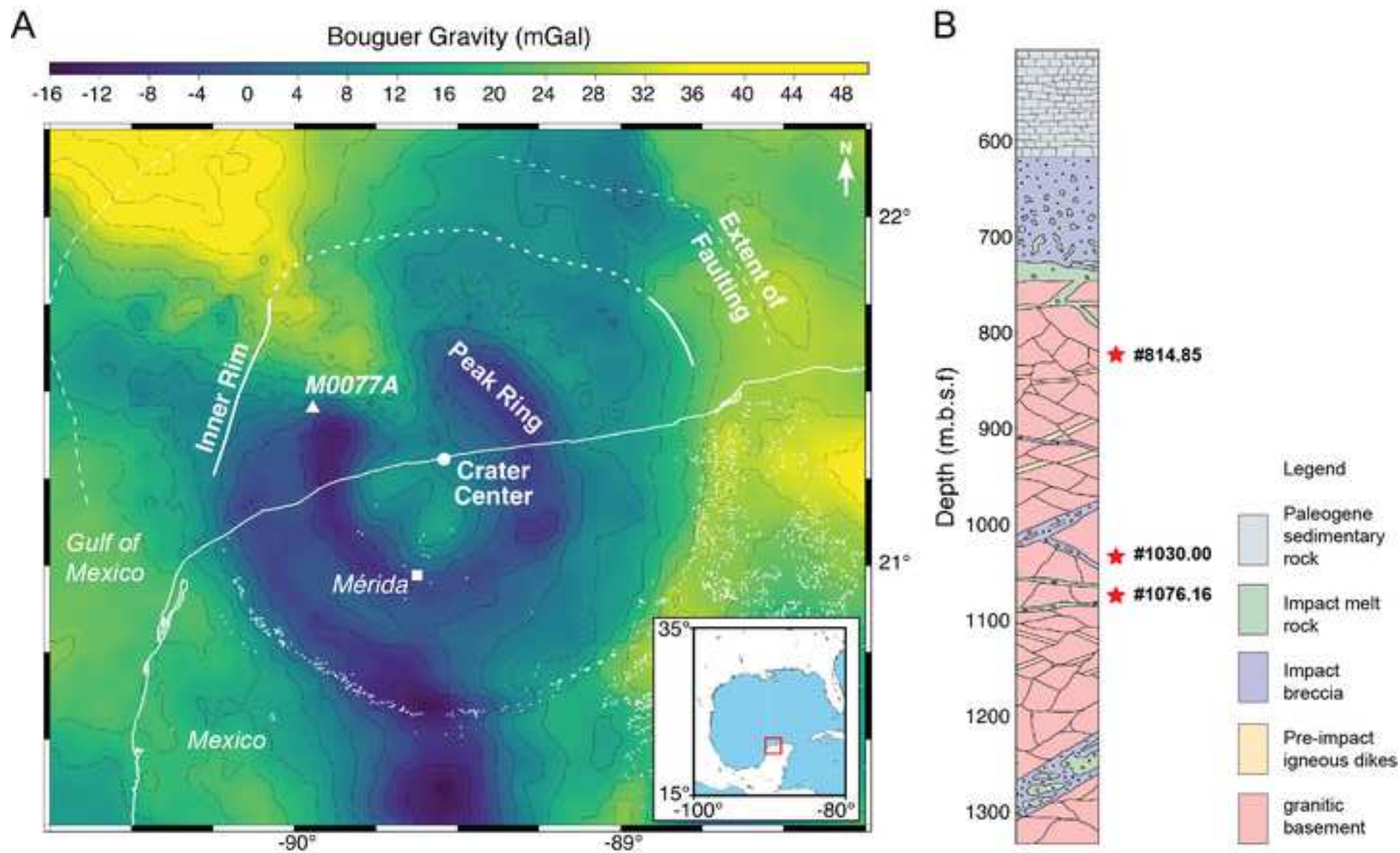
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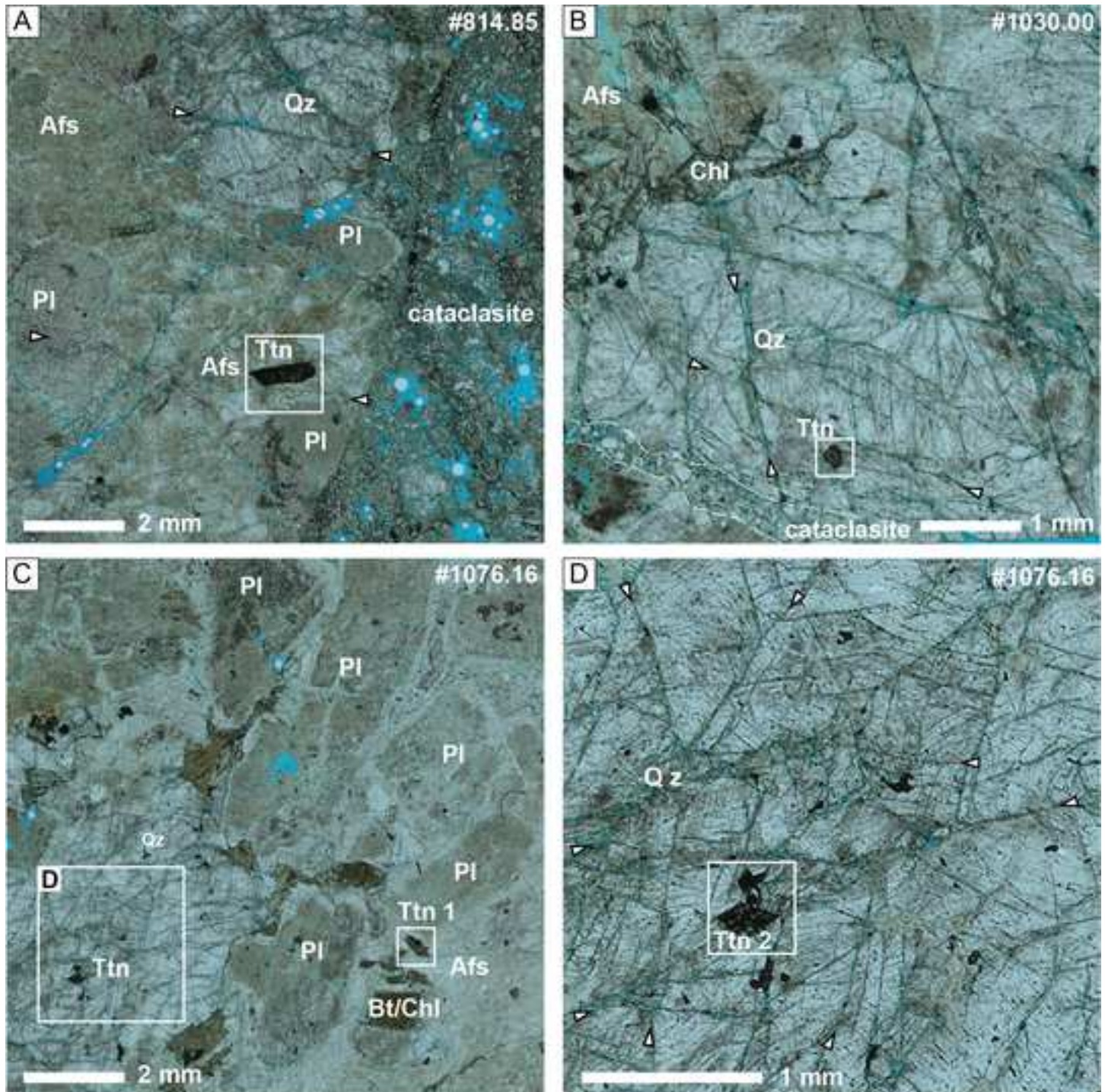
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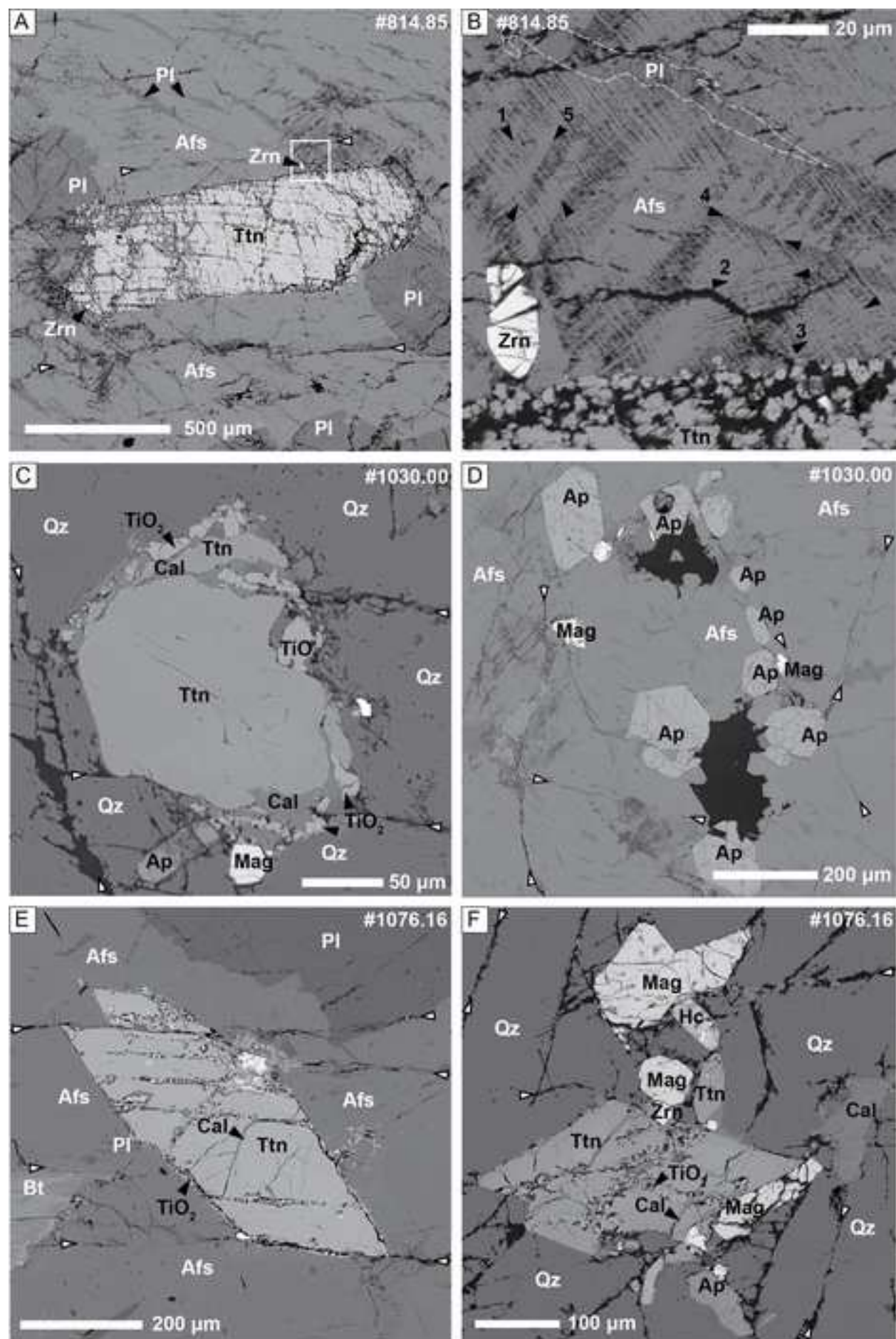
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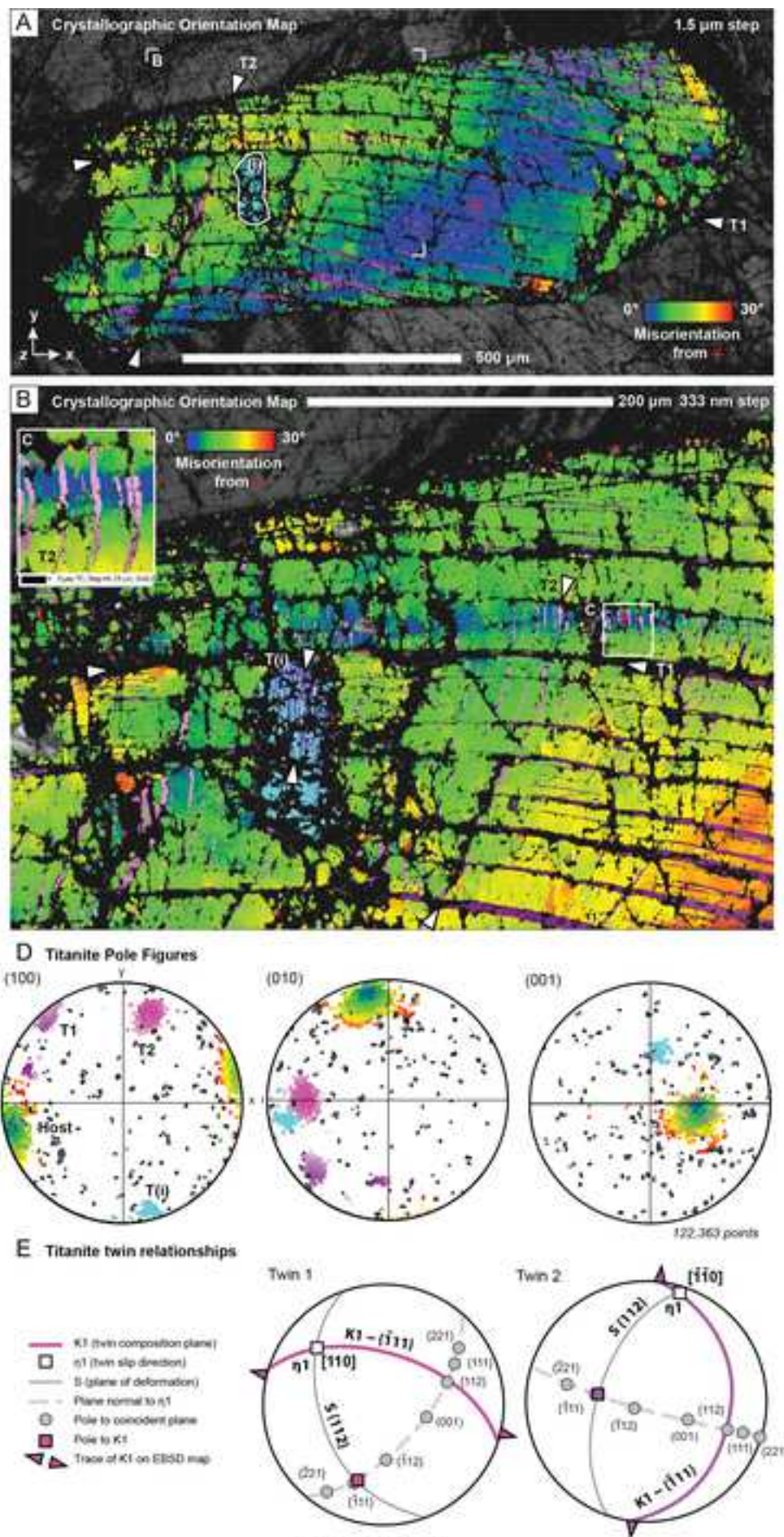
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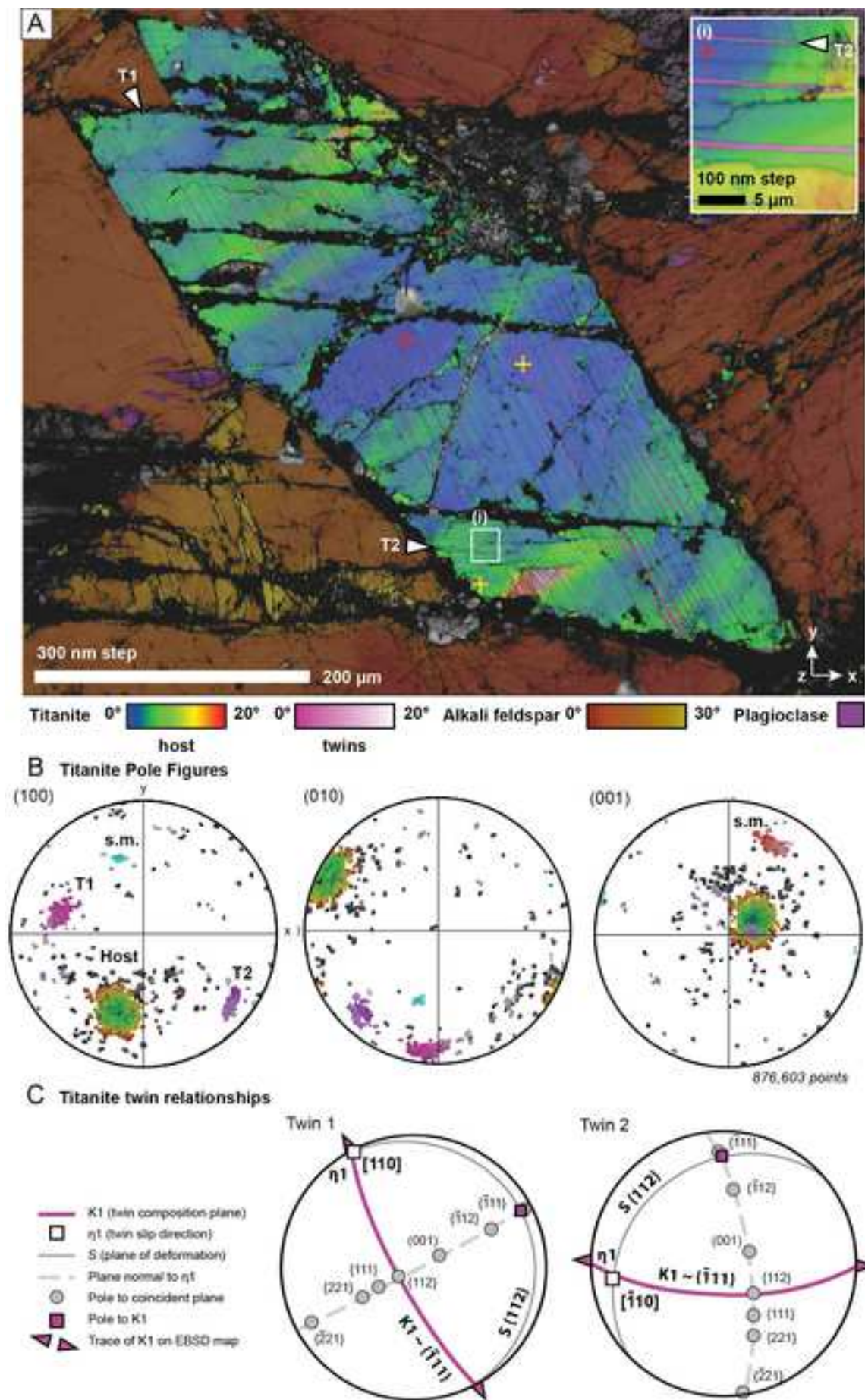
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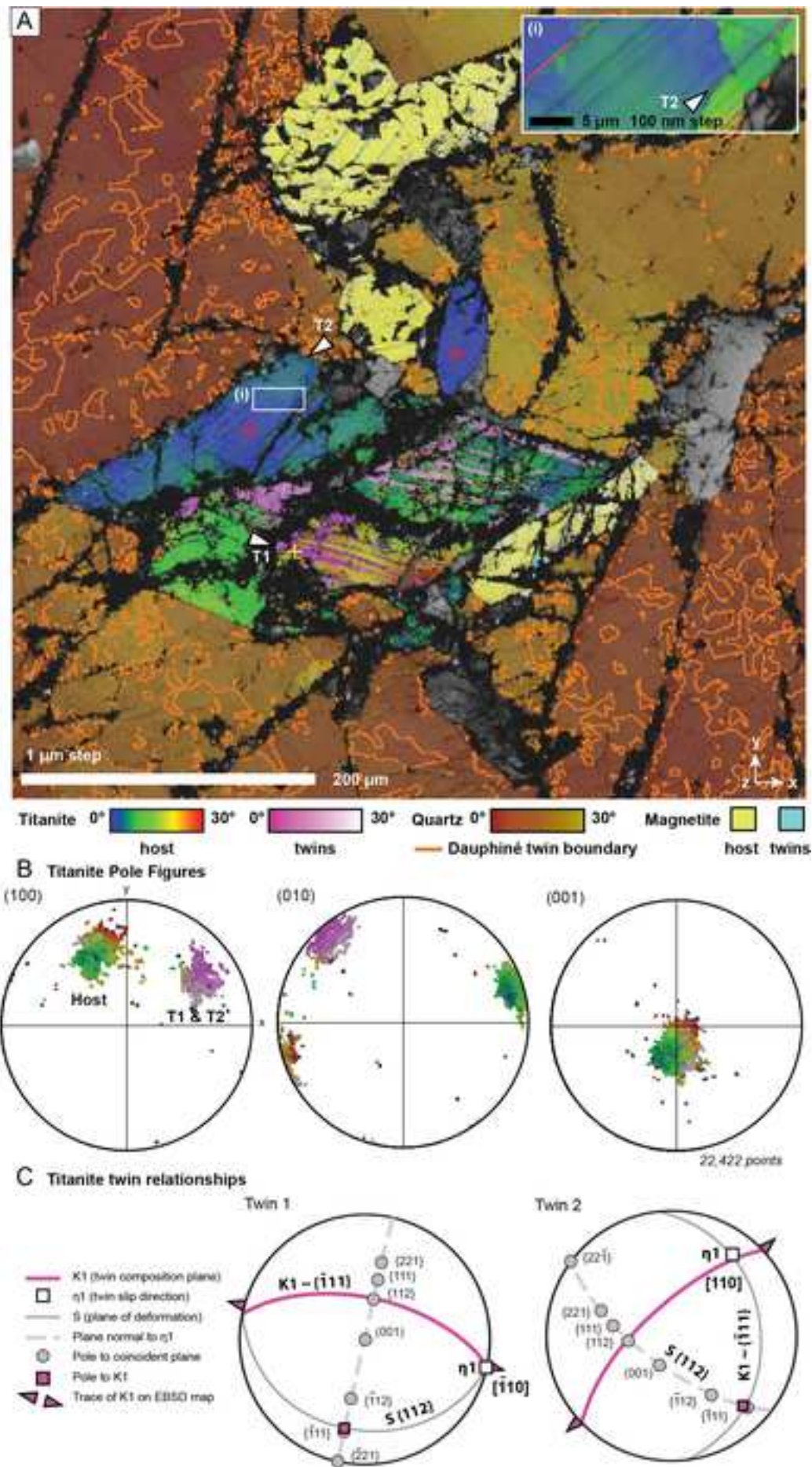
Timms Figure 3



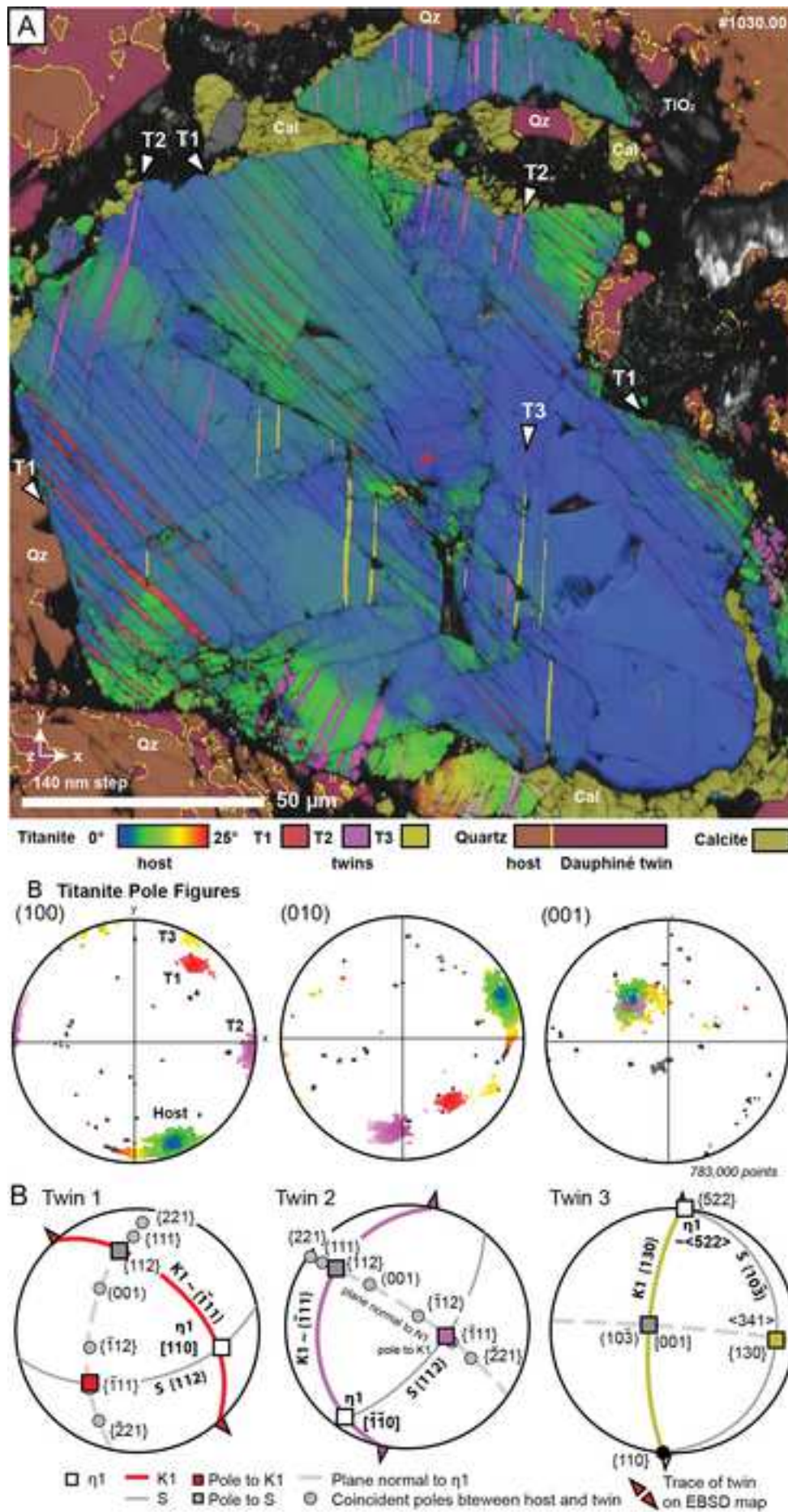
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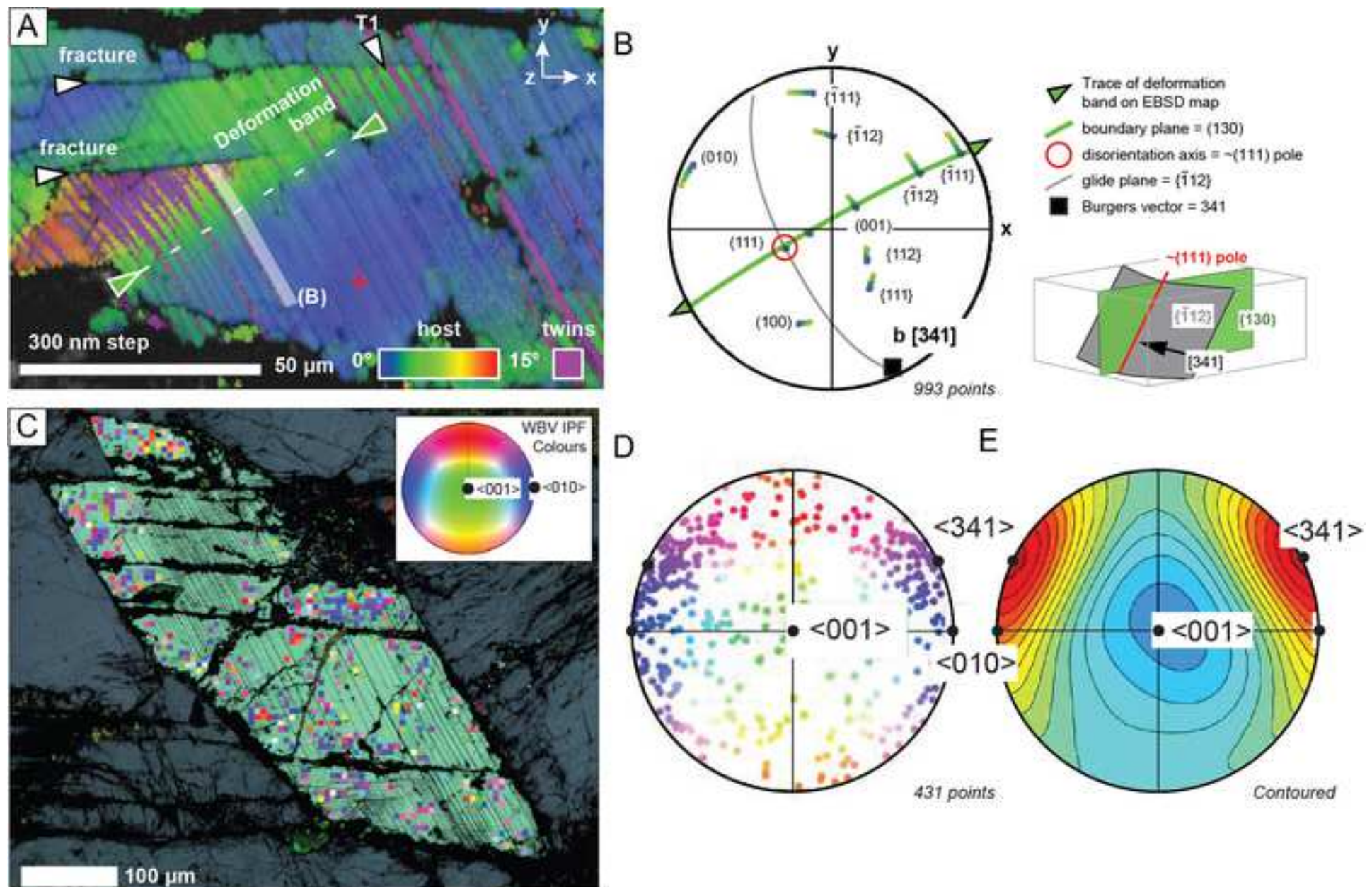
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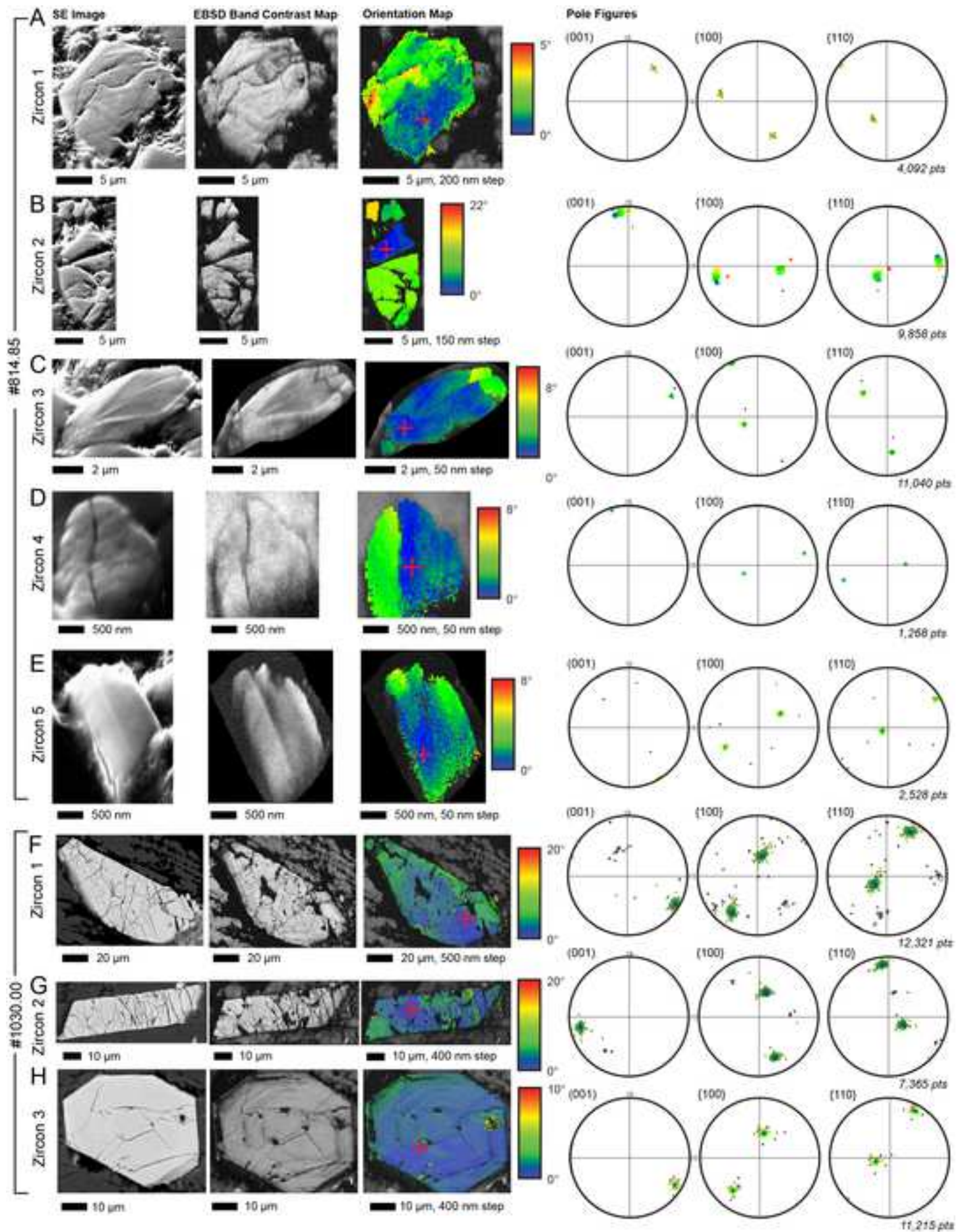
Timms Figure 6



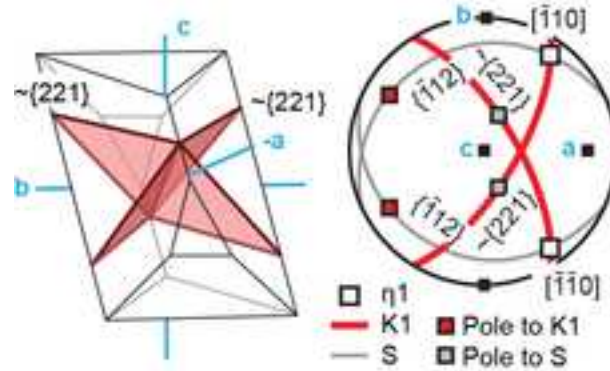
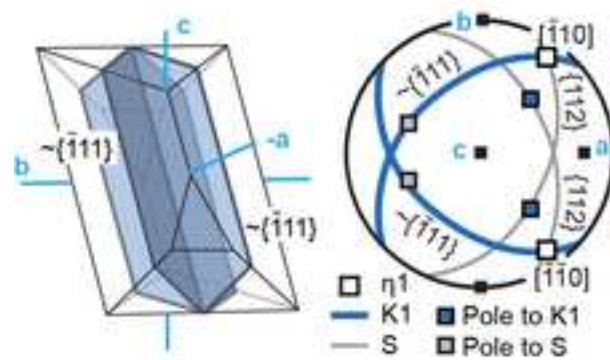
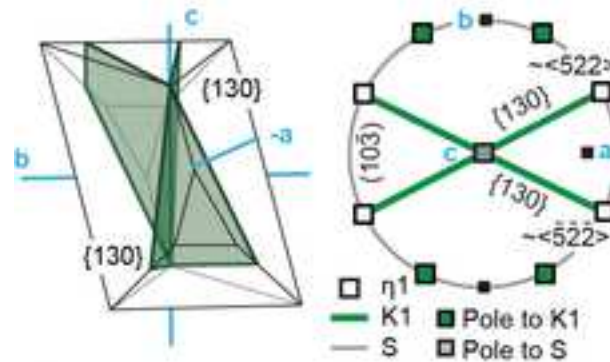
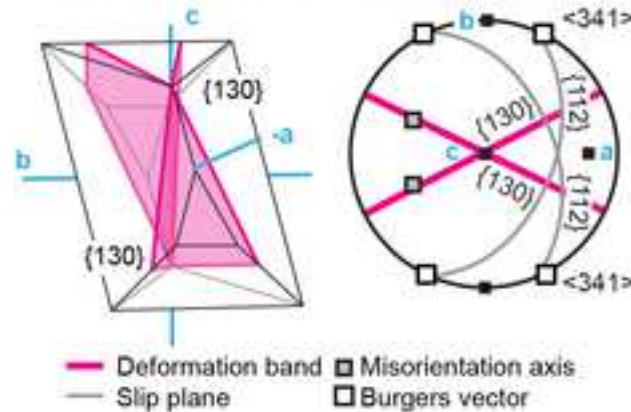
Timms Figure 7

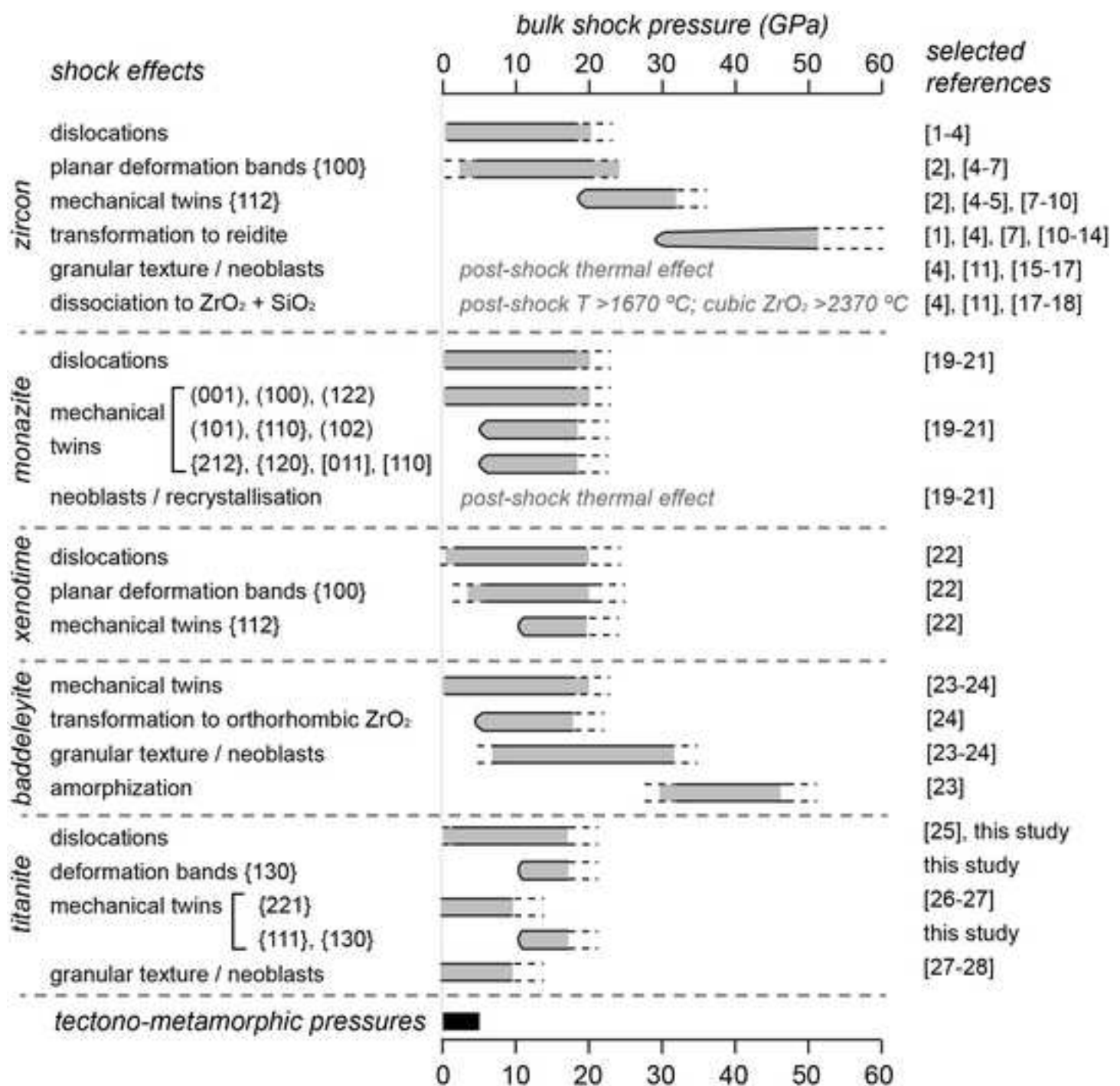


Timms Figure 8



Timms Figure 9

A $\sim\{221\} \langle 110 \rangle$ twins (Borg, 1970)

B $\sim\{111\} \langle 110 \rangle$ twins (this study)

C $\{130\} \sim\langle 522 \rangle$ twins (this study)

C $\{130\}$ deformation bands (this study)




Timms et al. Figure 11

Table 1. Details of samples used in this study. Sample IDs comprise information relating to the IODP catalogue system, and have the following naming convention: Expedition-Site-Hole-Core-Section-TopDepth-BottomDepth. Depths are driller depths given as metres below sea floor (m.b.s.f.).

IODP Sample ID	Depth (m.b.s.f.)	Brief thin section description
364-77-A-121-R-1-75-77	814.85	Shocked granitoid with thick cataclasite bands. Intact domains are dominated by plagioclase and alkali feldspar with perthite lamellae and local PDFs. Subordinate patches of quartz have abundant PFs and PDFs. Minor chlorite after biotite, and euhedral titanite. Accessory apatite, zircon and magnetite. Cataclasite contains patchy calcite cement.
364-77-A-204-R-1-7-9 (duplicate)	1030.00	Shocked granitoid dominated by large plagioclase laths and alkali feldspar with perthite lamellae, subordinate quartz patches with abundant PFs and PDFs. Minor biotite (partially altered to chlorite) and euhedral titanite. Accessory apatite, magnetite, and zircon. Conspicuous set of transgranular planar fractures are lined with calcite.
364-77-A-219-R-1-22-24	1076.16	Shocked granitoid with roughly equal proportions of plagioclase, alkali feldspar and quartz. Quartz contains abundant PFs and PDFs. Minor biotite (heterogeneously altered to chlorite), euhedral titanite and apatite. Accessory magnetite, zircon, and hercynite. Alkali feldspar contains perthite lamellae. Patchy calcite lines fractures.

Table 2. Scanning electron microscopy settings and electron backscatter diffraction analysis acquisition and processing parameters.

processing parameters.			
SEM			
Make/model		Tescan Mira3 FEG-SEM	
EBSD acquisition system		Oxford Instruments AZtec / Nordlys EBSD Detector	
EDX acquisition system		Oxford Instruments AZtec / XMax 20 mm SDD	
EBSD Processing software		Oxford Instruments Channel 5.10	
Acceleration Voltage (kV)		20	
Working Distance (mm)		18.5	
Tilt		70°	
EBSD match units			
Phase	Space Group	β (°)	
Titanite	15	113.93	American Mineralogist phase database (database family 725, best in family)
Quartz	152	n/a	'Quartznew', HKL database (Sands, 1969)
Orthoclase	12	116.07	American Mineralogist phase database (Prince et al., 1973)
Bytownite	2	115.87	ICSD phase database (Facchinelli et al., 1979)
Magnetite	227	n/a	HKL phase database (Wechsler et al., 1984)
Calcite	0	n/a	HKL phase database (calcite.cry)
Zircon	141	n/a	Zircon5260, 1 atm (Hazen and Finger, 1979)
Reidite	88	n/a	Reidite6032 0.69 GPa (Farnan et al., 2003)
Apatite	176	n/a	ICSD phase database (Gankev, 1996)
EBSP Acquisition, Indexing and Processing			
EBSP Acquisition Speed (Hz)	40	Band detection (min / max)	6 / 8
EBSP Background (frames)	64	Mean mean angular deviation (all phases)	<1°
EBSP Binning	4 x 4	Wildspike correction	Yes
EBSP Gain	High	Nearest neighbor zero solution extrapolation	8
Hough resolution	60		

Table 3. Unit cell parameters for titanite

A (Å)	6.554
B (Å)	8.708
C (Å)	7.069
α (°)	90
β (°)	113.93
γ (°)	90

Table 4. Host-twin disorientation and misorientation calculations for titanite

Grain and twin ID	Disorientation		180° Misorientation				
			Angle (°)	Best fit plane normal		Best-fit direction	
	Angle (°)	Axis <hkl>		Plane {hkl}	Deviation (°)	Axis {hkl}	Deviation (°)
121 T1	75.23	102	179.92	$\bar{1}\bar{1}0$	0.66	$24\bar{1}$	4.53
121 T2	72.27	$\bar{1}0\bar{2}$	179.33	$\bar{1}\bar{1}0$	1.59	$24\bar{1}$	2.60
121 T(i)	73.32	102	179.34	$\bar{1}\bar{1}0$	0.67	$24\bar{1}$	4.01
204 T1	74.16	102	179.98	110	0.23	$24\bar{1}$	4.12
204 T2	74.02	102	179.65	$\bar{1}\bar{1}0$	0.22	$24\bar{1}$	4.07
204 T3	51.30	001	179.97	$\bar{3}4\bar{1}$	2.11	$\bar{1}\bar{3}0$	0.20
219_1 T1	73.93	$\bar{1}0\bar{2}$	179.93	$\bar{1}\bar{1}0$	0.08	$24\bar{1}$	3.88
219_1 T2	74.17	$\bar{1}0\bar{2}$	176.85	$\bar{1}\bar{1}0$	0.73	$24\bar{1}$	4.37
219_2 T1	72.12	102	178.20	110	1.06	$24\bar{1}$	3.53
219_2 T2	72.58	102	179.98	$\bar{1}\bar{1}0$	0.75	$24\bar{1}$	3.18
Borg 1970	~73	102	180	110	n/a	221	n/a

Table 5. Twinning in titanite. Disorientation = angle/axis that describes the minimum misorientation between the twin and the host. 'Coincidence' is defined as $<0.7^\circ$ angular deviation.

Source Reference	Borg (1970)	Borg (1970)	This study	This study
Examples			#814.85 (T1, T2), #1030.00 (T1, T2), #1076.16_1 (T1, T2), #1076.16_2 (T1, T2)	#1030.00 (T3)
Twin Mode				
K ₁ (slip plane, plane of no distortion, composition plane)	Irrational $\sim\{221\}$	$\{\bar{1}31\}$	Irrational $\sim\{\bar{1}11\}$	$\{130\}$
η_1 (Slip direction)	$<110>$	Irrational	$<110>$	Irrational $\sim<522>$
K ₂ (Conjugate plane of no distortion)	$\{\bar{1}31\}$	Irrational $\sim\{221\}$	Rational ???	???
η_2 (axis of principal zone)	Irrational	$<110>$???	???
S (plane of deformation)	Irrational $\sim\{\bar{1}\bar{1}2\}$		$\{112\}$	($10\bar{3}$)
Twin type	2	1	2	1
Disorientation				
Angle	$\sim 74^\circ$	$\sim 74^\circ$	$\sim 74^\circ$	$\sim 53^\circ$
Axis	$<102>$	$<102>$	$<102>$	[003]
Coincident Planes				
(001)			Yes	No
$\{111\}$			Yes	No
$\{\bar{1}11\}$			Yes	No
$\{112\}$			Yes	No
$\{\bar{1}12\}$			Yes	No
$\{221\}$			Yes	No
$\{22\bar{1}\}$			Yes	No
$\{24\bar{1}\}$			(Yes?)	No
$\{130\}$			No	Yes
(103)			No	Yes
Coincident directions				
[001]			No	Yes
$<110>$			Yes	No
$<\bar{1}10>$			Yes	No
$<34\bar{1}>$			No	No